

Failure (How to know whether a component has failed)
A part or assembly is considered to have failed under any of the 3 conditions.

- Conditions
- When it becomes completely inoperable
 - When it is still operable but is no longer able to perform satisfactorily / doesn't give intended performance
 - When serious deterioration has made it unreliable or unsafe for continued use, thus necessitating its complete removal from service, safety is compromised

fundamental sources of failure :-

- Aspect of design.
- Material selection.
- Material imperfections.
- Fabrication and processing
 - Reaming
 - Assembly - error in assembly.
 - inspection.
 - testing
 - quality control.
- Storage & shipment -
 - Severe conditions
 - Maintenance
 - Chemical damage.

A Design.

- Mechanical notches -

- ① - Presence of mechanical notches at points of high stress.
- ② - The use of too sharp a fillet-radius at a change in cross section of a shaft.
- ③ - sharp corners in bending or torsional loading.

Tensile testing Specimen



→ No sharp corners.

Step.

- a) Material & fabrication
 - b) Macro examination of the fracture S/C.
 - c) Conformance to specification.
 - d) Micro examination of the S/C.
 - e) Analysis of crack origin.
 - f) Conclusion.
 - g) Corrective action.
- Insufficient design criteria.

- Impossibility for making stress calculations for complex parts.

- Insufficient considerations of the types and magnitudes of the loads to which the part will be exposed over service.

B - Selection of material.

- Inadequacy of tensile data.
- Specific characteristics of the material that quantitatively measure its resistance to failure.
- operating conditions. (temp, corrosion environment) etc.
- Mechanical properties changing with time.
 - a) Resistance to wear
 - b) Effect of elevated temperature on properties.
 - c) Resistance to stress corrosion, corrosion, corrosion fatigue and radiation.

C - Imperfections in Materials

→ internal & surface imperfections.

decrease the overall strength, provide preferential paths for the propagation of cracks, act as notches, serve as preferential pitting sites attract or provide path for intergranular corrosion.

Casting defects - Cold shuts, inclusions, porosity, voids, shrinkage cavities,

forgings -

Forging defects - laps, ~~seams~~, seams, shrinkage cavities, flow line pattern, segregation in the billet,

rolling defects -

D

D - Deficiencies in processing

Sometimes failures occur due to specification of unsuitable processing procedures, incomplete or ambiguous specifications, changes made in specifications without complete evaluation.

→ Cold forming & related operations.

Such as deep drawing, stretching, expanding, reducing and bending, produce residual stresses.

Surface defects and metallurgical changes caused by processing have influence on fatigue strength, resistance to brittle fracture, corrosion resistance, anisotropic properties.

- Shearing, blanking & piercing produce local residual stresses.

→ Machining & grinding

Machining introduces residual stress, severe grinding causes overheating and changes the microstructure.

→ Identification marking - by Marking is given impact-indentation, or by electro etching. - The indentation should not be in the highly stressed region.

- Identification mark by electrical discharge machining, produces flash and a heat-affected zone, which leads to the stress corrosion cracking.

→ Improper heat-treatment - - overheating, undertempering, use of low hardening temper. Excessive temperature gradients, use of quenching, tempering, aging conditions unsuitable for a specific alloy or part.

Decarburisation during heat treatment, induces failure by ^{a)} fatigue, because it decreases the endurance limit of the surface.

b) by distortion of small parts on which the strength is reduced due to decarburisation.

~~It is~~ Decarburisation is detrimental to the service life of springs and small shafts.

→ Acid pickling and electrolysis

(especially at low cathode efficiency) are well known for their ability to cause hydrogen charging and hydrogen embrittlement of high strength steels.

Chemical cleaning or electrolytic cleaning and etching in which hydrogen is generated produces similar effects.

Welding - Stress Corrosion Cracking of welded austenitic stainless steels of boilers, heat exchangers, and pressure vessels. Sensitisation & IGC may be reasons of failure.

- proper welding conditions must be maintained in order to avoid this. proper filler material to be used.

Reworking - Reworking occurs during some stage of manufacturing.

Errors in Assembly

- Not detected by the manufacturer or purchaser.
- Mostly associated with moving parts of mechanical assemblies or with electrical assemblies.
- But - structural components (for example small errors in the placement of rivet holes in airplane wings)
- operator negligence.

Improper Service Conditions

The operation of equipment - under abnormally service conditions of speed, load, temperature and chemical environment - without regularly scheduled maintenance, inspection and monitoring are the prime causes of failure.

- start up and shut down inspection should be proper. Most of the airplanes have crashed during take off or landing.

- Pitting Corrosion of austenitic stainless steel boilers during shutdown

General Procedure of Failure Analysis (HKB)

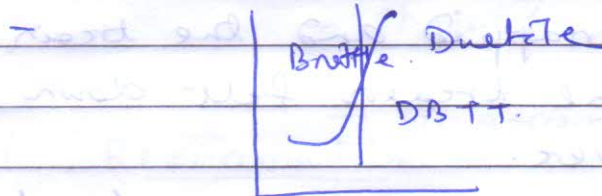
Guidelines to reach the root of the
cause of failure (Root Cause Analysis)

- Background information which provides what happened prior to and at the time of failure, the manufacturing history of the part and discussion with eye witness.
- Visual examination includes a visit to the scene of failure, making sketches, taking measurements, notes and photographs. It helps to select parts to be taken to the investigation agency, examination and understanding.
- Non-destructive examination to obtain defect-information on the surface as well as at the core of the failed part.
- Fractographic examination to identify the type and nature of fracture.
- Destructive testing to ensure soundness of the material in conformity with the specification in respect to chemistry, properties of microstructure, etc.
- Examination of all data to arrive at a conclusion of the cause of failure.
- Recommendations for its prevention.

Examples.

① Pittsburg - 3 M rivets. - impedance resistance of
 1912 binding the plates | the rivets were poor
 ↓ loss of 1500 lives. low quality iron

- 16 components were connected. at the end
 so sequential water filling took place



DBTT should be low.
 Addition of Ni, etc.
 use alloy steels.

Recommendations - Use alloying elements that -

- Avoid ~~Don't~~ use use of dissimilar metals
- Avoid use of poor rivets

② St. Francis dam - Los Angeles city (40 miles)

↓
 1928 Constructed in Canyon hills
 ↓
 dam failed → 1600 people killed
 ↓
 High water waves.

→ Regular check up of the dam

→ Reason - Failed due to residual stresses, poor construction, construction using low quality materials.

→ Recommendation - Regular check up.
 - Use good quality materials
 - Giving a larger factor of safety.

③ Pocomo Narrows Bridge Collapse (1940)

↓ suspension bridge in Washington.
 world's 3rd largest suspension bridge.
 longest

→ Bridge used to flutter with wind velocity

→ It collapsed at a wind speed (64 km/h)

→ The cause was aeroelastic flutter.

→ Avoid the suspension bridges.

Indian features

① - Kadakundi train disaster

- The railway bridge was around 140 yrs.
- One of the piers settled down the sand in river.
- As soon as the train came near it, the track snapped and the train derailed, the train fell down the river.

→ Recommendation → Never use a structure of high importance without testing it. Always keep an eye on it and test it.

② Bhopal gas disaster: Union Carbide.

1984, a toxic gas release at a union carbide pesticide plant - in Bhopal. resulted in 2259 deaths, immediate deaths and 10000 deaths following disaster.

558,128 injuries.

- It was caused by methyl isocyanate, a highly toxic and irritating material, in the making of pesticides, got contaminated with H_2O . This contamination is exothermic and released heat. The inside temperature raised to 200°C. This was beyond the capacity of the plant to sustain the temperature. Therefore, the emergency release systems kicked in, venting the extra pressure which escaped and began to spread.

The gas was heavier than air, it floated for miles.

③ Rafiqan: rail bridge collapse

→ Train fell down due to plate girder deterioration over time.

→ Bridge was rusty

→ Strength loss with use

→ lost strength of fatigue due

→ Girders were corroded

Recommendation - Make sure the design

is checked against fatigue and steel

is coated with anti-corrosion paints.

Mechanical Component Failure

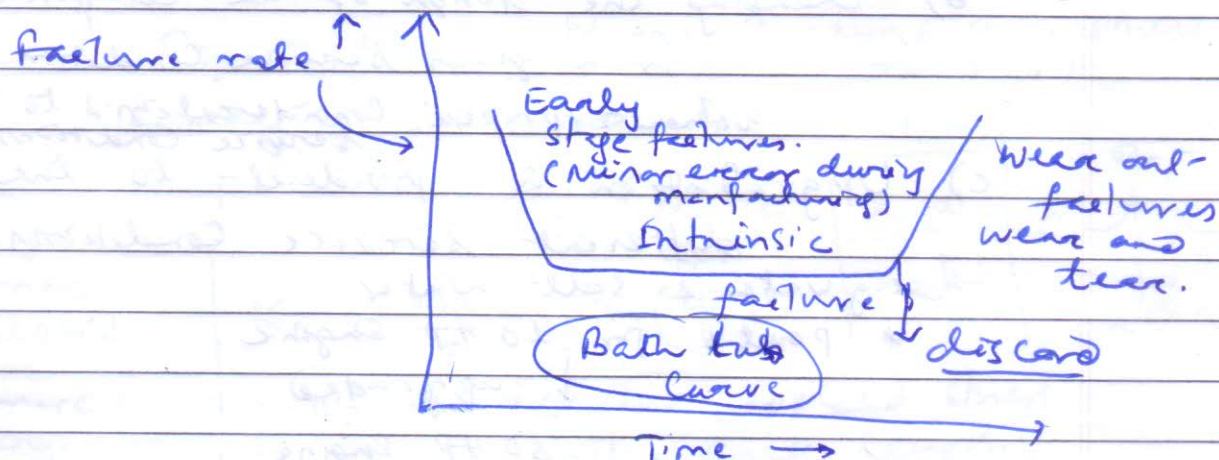
- Loading Condition.

① Elastic deformation (Acceptable)
Automotive shaft → crank shaft in automobiles.

② Plastic deformation (Unacceptable) - shaft - gears, teeth, twisted

③ Failure (D, B, F, E) → Creep, Fatigue
↓ Ductile fracture

④ Loss of dimension beyond acceptable limits, wear & tear, erosion, corrosion.



②

- ① Importance of FA
- ② What type of HW
- ③ What kind of expertise we need
- ④ How FA is useful for ensuring the improved reliability, quality and performance of the products

Fundamental sources of failure ③

- ① Improper design
- ② Improper selection of material
- ③ Imperfections in materials - inclusions
- 4 ④ Improper manufacturing processes - stress.
- 5 ⑤ Improper assembly - leads to imbalance misalignment -
- 6 ⑥ Improper service conditions
 Size & shape → too high temp.
 → loading condition
- 7 ⑦ Improper maintenance - little lubricants
 - Anti-corrosion
 Coatings.

① Deficiencies in design.

① - sizing & shape required

for the given service conditions.

a) presence of stress raisers

↳ Mechanical notches

b) Changing the design of the component for a given service condition.

without proper consideration to the service conditions.

c) Upgradation a product - to the different service conditions.

e.g. water & salt-water

* parts for 10 HP engine.

↓ upgraded

50 HP engine.

- Design is developed without full knowledge of stress conditions owing to complexity of geometry.
- Designing without proper assessment of service conditions (Load, Temp & environmental conditions)
- Inability to use proper criteria for designing the engineering components.
- Creep - fatigue - overload

a) presence of stress raisers \leftarrow Intentionally
 \leftarrow Unintentionally
 \downarrow imperfections

No stress raisers $\sigma = P/A$
 $\sigma_r = P/A$
 if some geometrical feature introduced

local's stress of stresses.

$\sigma_{max} = \frac{P}{A_s}$ \leftarrow stress concentration.


$\sigma_{max} > \sigma_{ref} \rightarrow \frac{\sigma_{max}}{\sigma_{ref}} = \text{Theoretical stress concentration factor}$

> 1


example - threads, holes, rivets.


σ_{ref} , reference stress (without any stress raisers)

Geometry of stress raisers.

Surface stress raisers are more dangerous than inside stress raisers	$\sigma_m = 2\sigma_0 \left(\frac{a}{r}\right)^{1/2}$	plate	
	$K_t = \frac{\sigma_m}{\sigma_0} = 2 \left(\frac{a}{r}\right)^{1/2}$	$\frac{x}{2ck}$	

$\sigma_0 \rightarrow$ is the applied stress.
 $a \rightarrow$ half crack length.
 $r \rightarrow$ Radius of curvature.


$P_t \downarrow$  stress coneⁿ is high.

$P_t \uparrow$  stress coneⁿ is low.

Important in case of brittle materials
Stress Concentration is high.

- To avoid - fracture. → Fillets, Grooves, holes, etc.

→ Size & shape of notches.

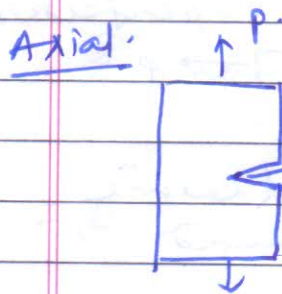
 but not on the size & shape of

use of
narrow
notches.

→ Type of loading - Axial
- Bending
- Fatigue

→ Metal aspects. - Torsional

assumed that material is
isotropic and material
behaves in linear elastic
material.



radius of crack tip.

P_t .

If material is brittle,
 $P_t = 0.5$, crack doesn't
blunt.

If m/l is ductile, the crack blunts
 P_t increases, stress
concentration gets reduced.

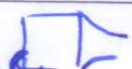
— In case of torsional loading, the
stresses are more at the surface
than at the core, implies that
torsional and bending stresses are
more harmful than the
axial loading.

→ High yield strength material are
more harmful.

→ The change in cross sectional
shouldn't be sharp but it should
be gradual.

gives
low
stress

Stress
concentration -
hole



Geometrical features (role of stress raisers)

- Design Criteria - fatigue, creep, Uniaxial.
- Type of Load - Axial,
- Material features - σ_y , σ_{UTS} .
- Metallurgical features - needles, platelets, lamellae.
(morphology of microconstituents) \rightarrow early nucleate the cracks.
- Calculate the maximum stress by taking into the stress raisers.
- fracture analysis - High strength, High hardness, Low ductility fracture mechanics. $K_{IC} = \sigma \sqrt{\pi c}$.

Case study - Leaf - 4. (Determination of design)

b) Changing / up gradation of a part to more severe condition.

- Unrotatability of the metal/alloy to the new application. \pm Good record at RT if we apply use it at low temp (-22°C) or Chemical industry (high temp), the behaviour will be different.

- Heat treatment requirements for steel - ~~tempered~~ quenched + ^{tempered}.

If the same part to be used at high temperature \rightarrow it will lead to over tempering \rightarrow loss in toughness.

- Improper stress relieving.
- Size/shape requirements for the new service condition. (high load), the size and notch of stress raisers play an important role. Loading distribution also changes in the new condition. Permissible stress may convert to torsional stress condition.
- Plane stress may change to plane strain.

- Upgradation of a product to a different-service condition

- Designing should be done as per the new condition.

These arise
↓
due to
Complexity
of
design.

- Stress distribution, Corrosion environment, Effect of environment
Improper distribution of load due to
Durability to service.

Calculate the stresses which will be acting on the components.

↓ material

② Improper Selection of Material - Selection of material for application

Each metal have different-

Physical & Mechanical properties

A $\sigma_{ys} \uparrow$ $\%EL \downarrow$

B $\sigma_{ys} \downarrow$ $\%EL \uparrow$ etc.

We need to consider the expected failure examination mechanism.

Double fracture

Brittle fracture

→ Creep

→ Fatigue

→ Corrosion

- Wear

- Erosion

- Wear.

~~The size and shape of the stress raisers.~~

- If failure is expected to occur by excessive plastic deformation at room temp. and high temperature, then the yield strength and creep respectively become important criterion of design.

- If failure is expected to occur by fracture under overloads, fluctuating loads, and impact loads then UTS, Endurance strength and impact strength respectively become the

design Criteria.

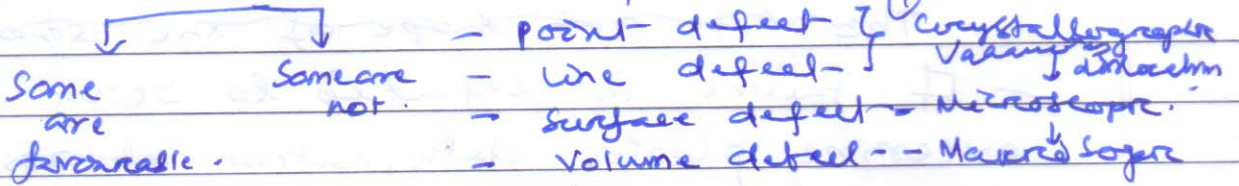
Failure Mechanisms / Selection Criteria.

Characteristics of materials required for the selection

Ductile fracture	-	yield strength (tension, compression)
Brittle	-	fracture toughness (K _{IC}) Notch toughness ductility DBTT.
fatigue	-	Endurance limit/ Fatigue strength with stress raiser, hardness
Thermal fatigue	-	Ductility, plastic strain. (under operating condition)
Creep	-	Creep rate at given temperature
Plastic deformation (change in cross section)	-	yield strength.
SCC (tensile load under corrosive environment)	-	K _{ISCC} , Corrosion resistance to specific environment

Imperfections in base metal itself.

- * perfect metal & Alloy not possible
- * live with imperfections



Point defects (Vacancies) → Vacancy

$$\frac{n}{N} = e^{-H_f/RT}$$

- interstitial impurity
- self-interstitial
- substitutional impurity
- Frankel / Schottky defect

Dislocations - 1 + edge
T - edge
screw
2 G

Some photos
in
localised way.

- ② Microscopic - bands, segregation, Unfavourable grain orientation, Grain boundary / Orientation.
 ↓ direction of loading / grains are unfavourable oriented.
 high low.
 [Diagram showing a rectangular grain with a horizontal line through it, representing a grain boundary or orientation.]
- ③ Macroscopic or Volume defect - Pores, Inclusions, Cracks.

Cracks at - SIC and internal cracks in the form of pores, inclusions.

In Al-Si alloy Si exists in the form of inclusions.



- Area gets reduced.
- Nominal stress increased.
- Act as stress concentrations. (tip radius)

Due to presence of defects, the Area ↓ stress ↑.

Lower radius, → higher is the stress concentration.

② Favourable engineering

Thermal & Electrical resistivity.

- Diffusion
- Dislocation density → increases the strength & hardness. → yield strength increases.
- GB strengthening: → Electrical
- Fe - Cr, Ni, Cr, Mo. Al - Mg, Si

Micro - Not detected. - NDT.

Macro - Detected by MDT.

Origin of Microscopic imperfection -

- Dissolved unfavourable gases H_2, N_2, O_2 etc.
- Inclusions of oxides nitrides, slag, Al
- High aspect-ratio micro-constituents.
- Heterogeneity in chemical composition. e.g. Segregation or selective depletion of few elements

- Varying metallurgical structures due to varying thermal cycles and deformation by metal during its production
- Unfavourable morphology - banded, laminated, plates, needle shape structures.

Origin.

Macroscopic features

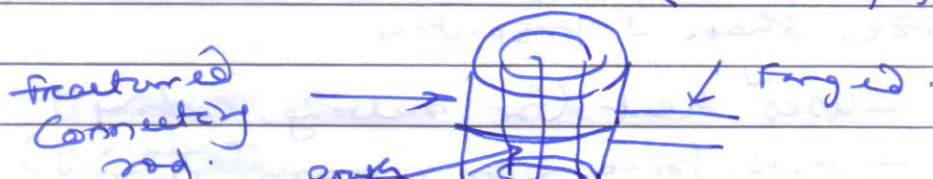
- Ingot of base metal: Cold shut, inclusion, pores, shrinkage voids, cracks (hot/cold)
- Forging: laps, seams, shrinkage, cavities, flowline pattern.
- These features may be left in base metal due to improper quality control in place.
 - S/C defects - cracks, open holes, blisters, laps, etc
 - Sub-surface - Blow holes, pores, piping defects, cold shut, lack penetration.
- These macroscopic features increase the stress and cause failure.

Effect of Imperfections.

- Micro-constituents
 - Varying mechanical properties (due to localised hardening/softening)
 - Reduce tolerance to crack nucleation (for hard and brittle metals)
 - Sources of stress concentration as notches e.g. banded structure, needles and laminates.
 - Delayed cracks (due to dissolved hydrogen in steel)
 - Reduced corrosion resistance due to increased galvanic cell formation tendency like in castings owing to segregation.

Effect ofCase-study (forged Connecting rod).

- Medium Carbon steel Connecting rod
50% P, 50% S: failed
- Composition within limits
- No metallurgical superfections
- Hardness was 140 BHN instead of
160-205 BHN.
- Metallography - Unweld showed homogeneous
pearlite with equal F/P showed proper HT.
- failed one showed banded structure
near feed marks.
(bands of ferrite)



- Fracture by fatigue - beach structure.
- The banded structure fell at
the high stress zone.

→ fracture started from transition in
Cross section.

- Beach marks indicated fatigue fracture
with smooth region of crack initiation.
- No plastic deformation - secondary
fatigue fracture.
- Die-pinchmark-test showed fatigue
cracks.

Reasons - Pending

- Notch sensitive banded structure in high
stress area - leads to nucleation & growth
- Rough ground marks stress raiser ^{of cracks}
- Low hardness at the crack initiation
site.

Recommendation: - Control of microstructure hardness and to finish in critical areas.
Uniform - Uniform microstructure to avoid stress concentration.



Failures caused by Improper Manufacturing Process/Conditioning

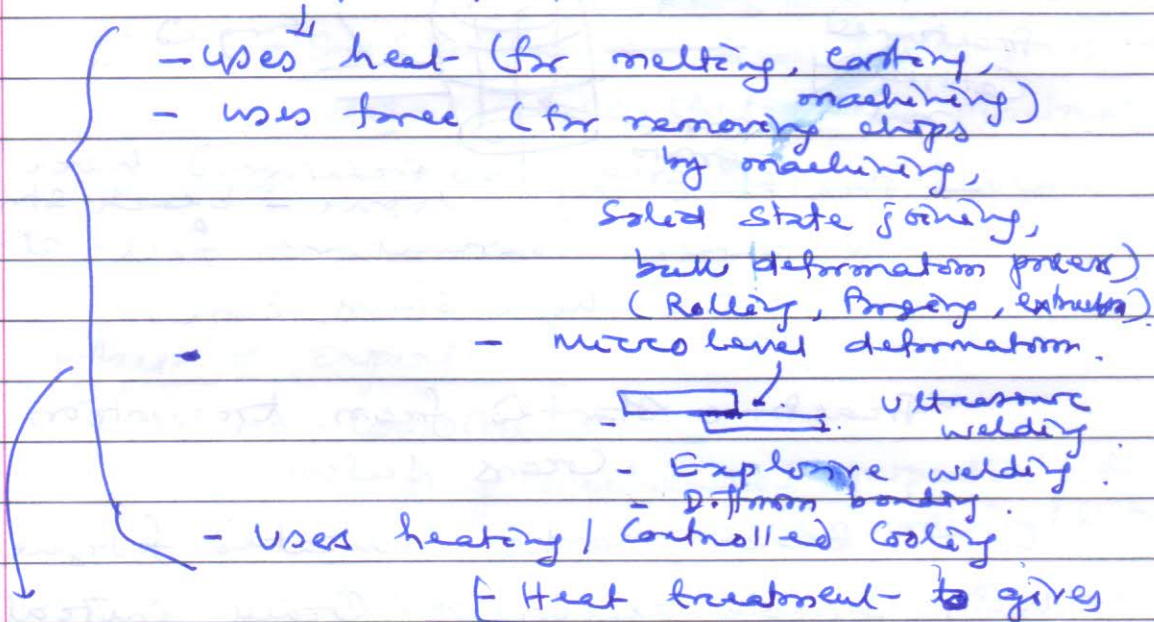
Improper Manufacturing leads to failure

- General factors
- Specific factors to each process

General factors

- Proper procedure not followed
- General procedure must be following

Size, shape & properties.



Taking these factors

into consideration.

the manufacturing process is established.

Example - Casting

Melt → Melt treatment - pouring

- solidification - cleaning

Example Welding (fusion welding)

Edge preparation + cleaning +

Substrate welding process (Gas or shielded metal arc) + Substrate process parameters + chipping + clean.

Process → Process Parameters, Secondary treatment

These should be very specific and strictly followed to get desired shape, size & properties.

- General factors / General factors to determine
- Unsuitable specifications in primary procedure.
 - Suitable melting facility
 - Process parameters
 - Incomplete + Non clarity in specification.
 - changes in specification without proper evaluation of the conditions.
 - failure to follow the specified procedure.
 - Inadvertent error on the part of worker.
 - Accidental damage.

If General factors are not followed, it can lead to -

- Discontinuity of product
- Large defects (cracks, pores, blowholes etc.)
- Unfavourable properties.
- Unfavourable metallurgical changes - softening/hardening

Effects.

These lead to.

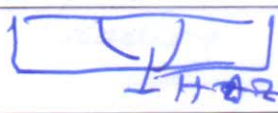
- Stress raisers
- ↓ Mechanical properties.
- ↑ sensitivity to cracking.

Manufacturing and performance are related.

Castings - As-cast - low in strength, ductility, toughness & fatigue strength (dentrite)

Welding - partial melting (fusion)

Leads to: residual stresses, variability in mechanical properties, distortion



Machining - Mechanical force applied by shearing. Thin deformed layer at surface \rightarrow \uparrow TS \uparrow fatigue strength.

Forming - Deformation is accompanied by work hardening. \downarrow ductility \downarrow Toughness
(Low temperature) \uparrow strength \uparrow Hardness.

If deformation is carried out at high temperature, without much adverse effect on mechanical properties.

Specific factors

\hookrightarrow Casting, welding, machining, forming processes.
(deformation)

Casting - * Metal + gases \rightarrow slag.
(dendritic structure)

\downarrow
flux.

\downarrow

Slag inclusion
Flux inclusion.

* Pores.

* Alloy segregation tendency.

- \downarrow
- Metallurgical Variations
 - Galvanic cell
 - Corrosion tendency
 - Solidification cracking
 - Corrosion

* Non uniform cooling - Variation

in micro structure.
 \rightarrow High hardenability
(hot spots)



shrinkage cavity

it is avoided by

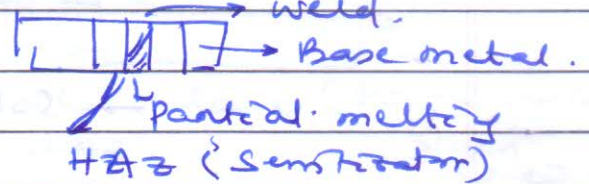
directional solidification

(chills / chaplets / riser)

Castings - Poor dendritic structure

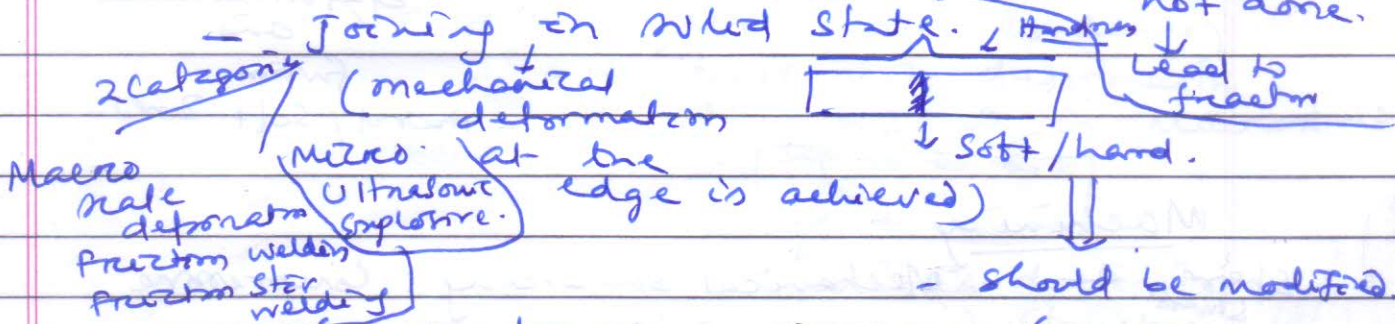
- Poor in hardness, strength, toughness & ductility.
- Scope for internal defects
- Heterogeneity in composition & property

Welding - Edges are brought to molten state.



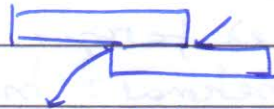
→ If procedure is not followed.

post weld heat treatment - is not done.



- should be modified

Solder - liquid based process (brass & silver)



Joints are weak
Can not take high load & high temp.

useful for electronic circuit which doesn't require high temp & high load.

Liquation

- Liquation cracking (cracking next to fusion boundary).

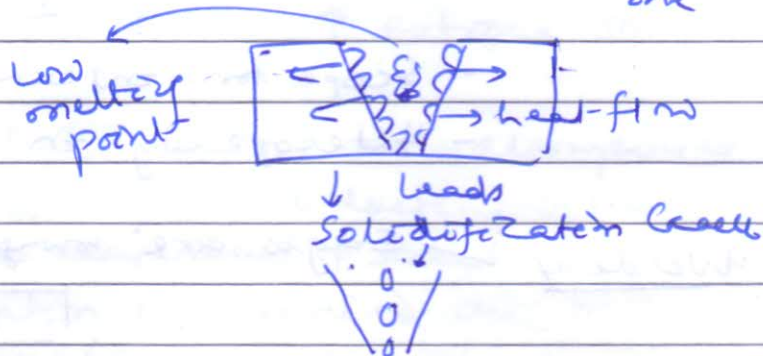
- HAZ.

- H₂ in HAZ (Hydrogen induced cracking)

- Gases (O₂, N₂) promote inclusion & delayed cracking.

- Stress corrosion cracking.

→ Segregation of alloying elements at the centre of fusion zone



→ Solid state joining.

- chelation.
- limited metallurgical bonding due to lack of deformation and fusion.
- Hard / soft zones

Machining

Shear stress

← Mechanical shearing: Compressive residual stress at the surface so improved fatigue and tensile properties.

- a) Laser beam
 - b) Flame cutting
 - c) Electron beam
 - d) Plasma
 - e) Electric discharge
- Thermal: melting and ablation.
 re-cast layer, heat affected zone
 may cause abrupt change in properties as per metal systems.

HAZ

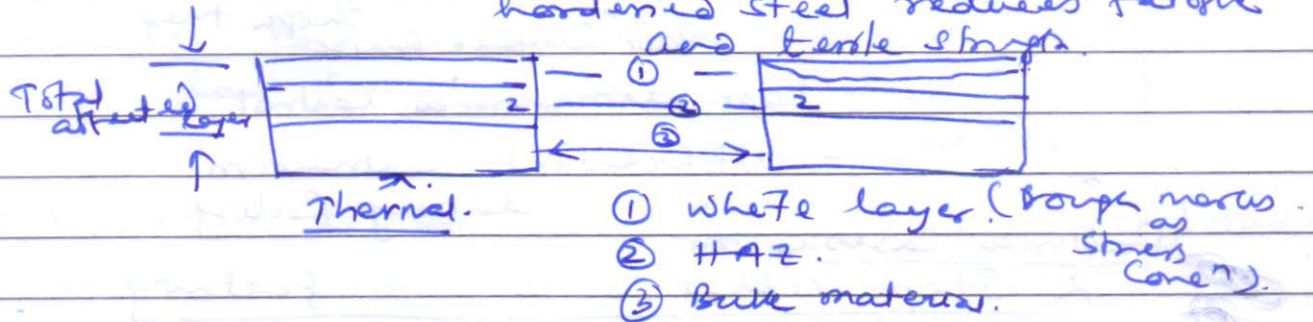
↓ 2v
the
near surface
layer.

- Chemical: Electrolytic - No residual stress so lower fatigue and tensile properties w.r.t sheared surfaces

No deformed layer
No residual stress.

failure of machined surfaces.

- Irregular surfaces and feed marks.
- Hard Spot formation.
- Localized softening.
- Cast layer with cracks with hardened steel reduces fatigue and tensile strength.



Forming:

- Hot (Strength by grain refinement)
- Cold (Strength by work hardening)
 - ↳ Non uniformity deformation
 - ↳ Non uniformity in microstructure



- decreased ductility.
- banded structure.
- defects.

failures in deformation based processes.

Cold forming

- Reduced toughness
- Increased notch sensitivity
- Surface cracking (material)
- Unfavourable grain flow pattern, stronger in longitudinal direction.
- These reduce fatigue, corrosion and increase brittle fracture resistance.
- Residual stress due to localized differential deformation :-
 - anisotropic mechanical properties
 - localised loss of ductility
- Increased DBTT (steel)

Hot forming

- Doesn't cause much problem.
- They gain strength from the grain refinement.
- Oxidation tendency ^{become} at high temp.
 ↓
 Poor surface finish.
- Poor dimensional control.
- Differential contraction during cooling.

Defects caused by2 Steps related to manufacturing

① bench marks. Indentation

② ↑ Mechanical properties.

(Heat ^{improper} treatment conditions to ^{remove} ~~remove~~ residual stresses)

① Indentation Marks

① - Mechanical based

Indentation (using punches)
 ↓
 Heavy section.

Embossing
 ↓
 Sheet metal.

② - Electrochemical Etching. - controlled removal material.
 (Numbers / letters)

Mechanical based

- Displacement of metal ^{sidewise} caused by the plastic deformation of material ^{by punch}.

Displaced material. (hardened material)

If they are present - at the high stress region (e.g. shaft)

↓
 they will act as stress raiser

↓
 early nucleate cracks.
 (e.g. indentation at slc of shaft)

due to
2 factors

Stress Concentration.
Hardening of metal.

They will lead to
localised necking
& growth of cracks

e.g. Shafts and Springs at the surfaces where surface stresses are higher.

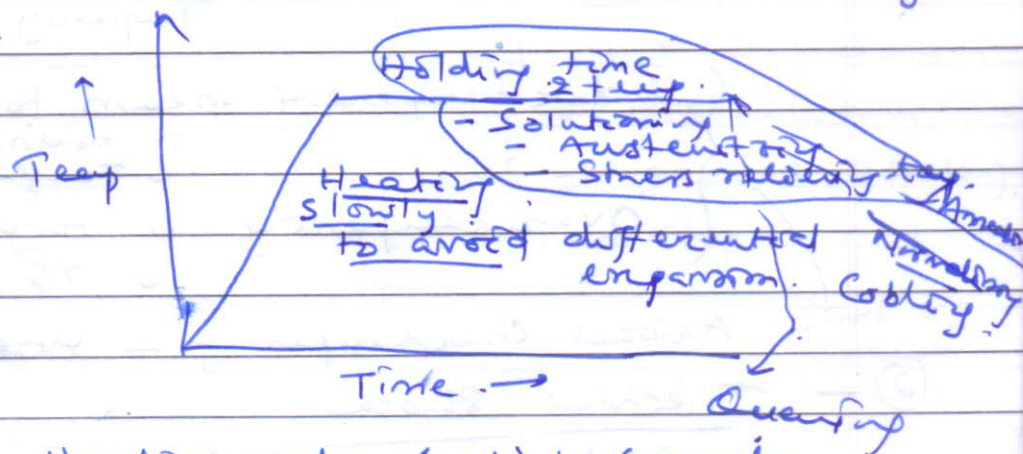
- Stress concentrations are ~~not~~ more harmful in dynamic loading (Fatigue or impact loading).

Failure due to ~~slow~~ indentation are more important in HAZ and hardened steel ($HRC > 50$) because these materials have low ductility (don't give localised yielding) and hence no stress relief.

② HEAT TREATMENT

1. Normal heat treatment - mechanical - Control of structure.
 - * ↓ Residual stresses
 - * ↑ stability in terms of properties / Uniformity / dimensional stability.

Procedure



- Heating rate should be correct
- Holding time "
- Holding temp "
- Cooling rate "

Inappropriate in any of these factors will lead to failure.

Defects in heat treatment -

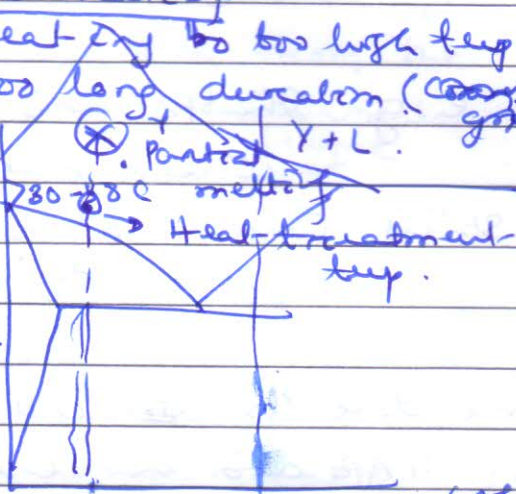
① Over heating

- a) Partial melting of low mp phase
- b) Coarse grain
- c) Oxidation
- a) Heat temp too high temp.
- b) too long duration (Coarse grain)

a) De Carbonisation

↓ Hardness
↓ yield strength

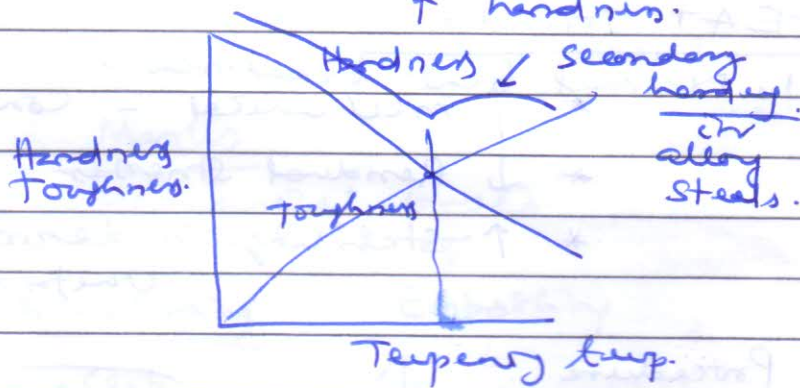
b) Spheroidization Fe₃C leads to softening.



1.2% C. (At low temp it will undergo Partial melting)

a) Prolonged heating

Quenched steel. ↓ toughness.
↑ hardness.



Under tempering means to overtemper

↑ Hardness.
↓ Toughness.

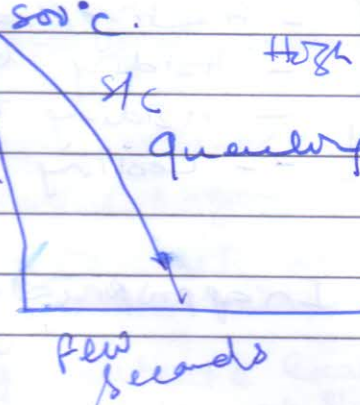
Over tempering ↓ Hardness
↓ TS.

Avoid Over tempering & Under tempering.

② Thermal shock

a) Fast quenching 500°C.

Quenching leads to quench cracks.



High temp crack gradient at 5% & low will lead to quenching cracks.

b) Suitable of the type of heat treatment.

A, N, O, tempering, Austenitizing
temp.

- High

~~High~~

- Improper Assembly

Various part parts are brought together & joined by mechanical or thermal.

↓
brazing
soldering
welding.

- Proper position & they should maintain ^{constant} the relative movement.

- Main causes are

- Ambiguity in assembling / Not clear
- Proper procedure to be followed.

- Insufficient or improper practice procedure.

- Misalignment - (& imbalance the force distribution
e.g. gear & shaft)

- ~~Very~~ Poor workmanship.

- Accidental error on part of operator (inadvertent error)

These are to be avoided.

A proper procedure to be followed.

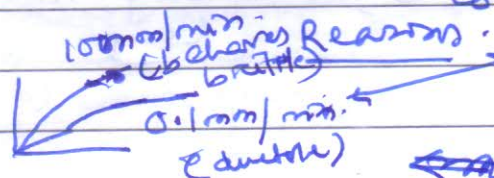
- Example - Tightening of Car wheels.

Tightening of nuts are done in a particular sequence, how much torque is to be applied for nuts.

- Example - Gear teeth flatness.

Improper Service Conditions.

Lots of variables,



- Rate of loading
- Type of force
- Magnitude of load
- Temp. (Normal, low, high)
- ~~unfavorable~~ Environment
(Dry, Moist, Seawater
Corrosive environment)
- Wear (Adhesive, Abrasive, Erosion, Cavitation)

Based on
Optimum Estimated failure
mechanism.

Substrate
material
& design features

Example

at σ & TT - 450°C \rightarrow Service temp 300°C
 of Service temp 450°C

\downarrow
 it will cause
 Over tempering.

- Mild steel weld joint - upto -20°C

if temp $< -20^\circ\text{C}$

it will cause brittle
 fracture.

- Titanic. failure. (Change in temperature
 conditions)

- Normal inspection & fracture maintenance

- Substrate repair is to be done.

Example

Boiler tubes \rightarrow start ups and shut downs.

✓ Start up is critical for a
 component

✓ Pitting corrosion in super heated
 tubes of boiler during several
 annually shut down.

(50mm x 4.12mm)

~~was~~ - The boiler tubes were operating at 400°C and 650 PSI for 2 years.

- The life period was 15 years.

- During hydrostatic testing during maintenance, it failed. tube

Investigation

- hot water - ~~the~~

Medium carbon

staying material - ~~not~~ steel. steel

* Visual - black adherent - magnetic scale with ^(internal SIC) localized presence of powdery, loose non-magnetic black scale showed pits like on removal.

* Analysis of deposit. XRD showed

Fe₃O₄ (black iron oxides), Fe₂O₃ (red dirt), FeO (without-chlorides)

* Metallography - Normal medium carbon steel structure, no segregation of in ~~micro~~ the microstructure.

Analysis

- Absence of chloride showed no HCL effect - used for cleaning of tubes. ~~was~~

- Fe₃O₄ formation suggest presence of little oxygen for pitting.

- Fe₂O₃ from loose, powdery non-magnetic brown deposits suggests pitting caused by presence of excess oxygen.

Observed in normal moist condition.

- During shut down of 15 days annual overhaul for requisite repairs.

- Oxygen rich condensate caused pitting in low points bend regions.

during overhaul period

CO₂ desolved water gets

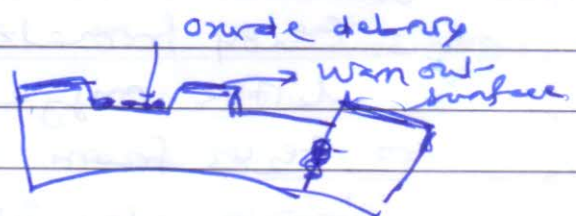
Condensed at the lower point bend region during overhaul

Improper maintenance

- A well developed and clear maintenance plan with steps for maintenance, including what, when, where, who and how, is specified explicitly (very clearly)
- Lack of information on proper schedule of maintenance, procedure of the maintenance frequently causes premature failure of moving components (where lubrication is important)

§ Example Splined end of Y350 steel drive shaft

- failed after 701 - 700 hr.
- shaft of Y350 hardness. HRC 34-38
- Case hardened to HRC 48-52 minimum depth 780 micron.
- Counter surface C58-62 with MoS_2 coating



Observation

1. Oxidized debris at the root area
2. No residual lubricant on splined shaft
3. MPT showed no surface defects
4. Chemistry & metallurgy as expected

Conclusion - failed due to excessive wear due to breaking down of lubricating film.

Recommendation - lubrication after each 300 hrs and hardness increased from 48-53 to 58-63 RC.

Toughness & Fracture Mechanics.

Toughness is the capacity of a material to absorb energy by deforming plastically before fracture. It is determined by the combined strength and ductility of a material and toughness is measured by the amount of work absorbed during the propagation of a crack through a structural member or a standard specimen.

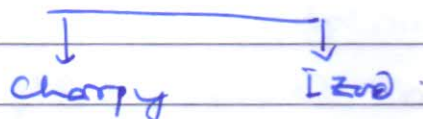
The consideration of toughness during design

1. permit selection of materials with low probability of failure by fracture.
2. Establishing the conditions for failure.

Toughness measurement are done by 2 types.

1. Standard tensile specimens :- Area under a standard tension stress-strain curve taken to fracture.
2. Notched or pre-cracked :- Measured by a fracture of pre-cracked specimens to measure the resistance to rapid crack propagation.

Notch toughness - Measured by testing a prescribed notched-bar specimen at known temperatures in a single blow pendulum type impact machine.



Notch toughness measurement by Charpy & Izod developed during World War-2 due to failure of 250 welded transport ships.

19 of which broke into 2 pieces.

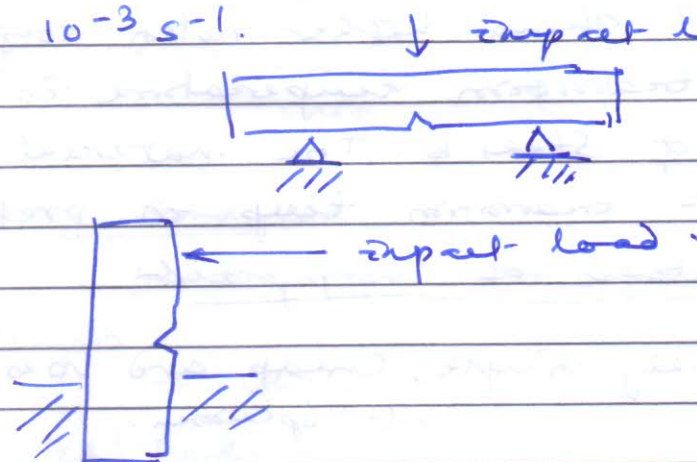
Charpy

Square Cross Section - $10\text{ mm} \times 10\text{ mm}$.

45° V-notch.

2mm deep with a 0.25 mm root

The specimen is supported by a radius. as a beam in horizontal position and loaded behind the notch by impact of a heavy swinging pendulum at impact velocity 5 m s^{-1} . The specimen is forced to bend and fracture at a high strain rate 10^{-3} s^{-1} .



The specimen is thick enough to ensure a high plan-strain loading & triaxiality stress.

The standard Charpy V-notch specimen provides a severe test for brittle fracture.

Information from Notch toughness

→ Measures the energy absorbed in fracturing the specimen. After breaking, the test bar, the pendulum rebounds to a height, which decreases as the energy absorbed in fracture increases. Expressed in Joules.

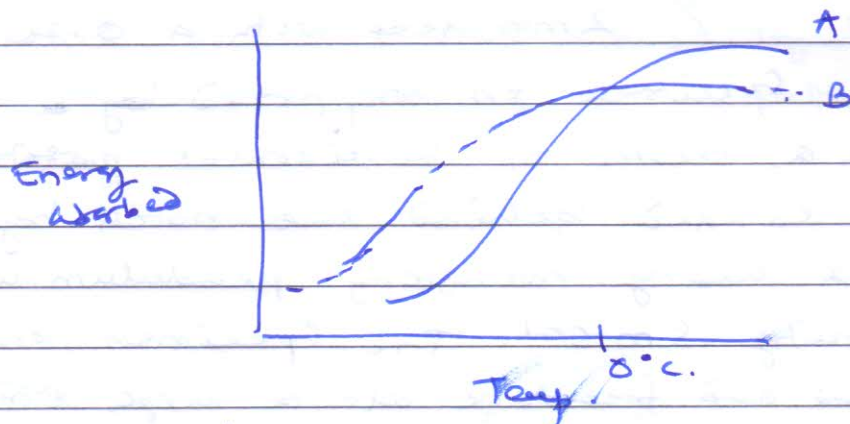
→ Examination of fracture surface

Fibrous (Shear fracture) Ductile

Granular (Cleavage fracture) High reflectivity (Brittle)

- Ductility - Indicated by percentage contraction of the specimen at the notch.

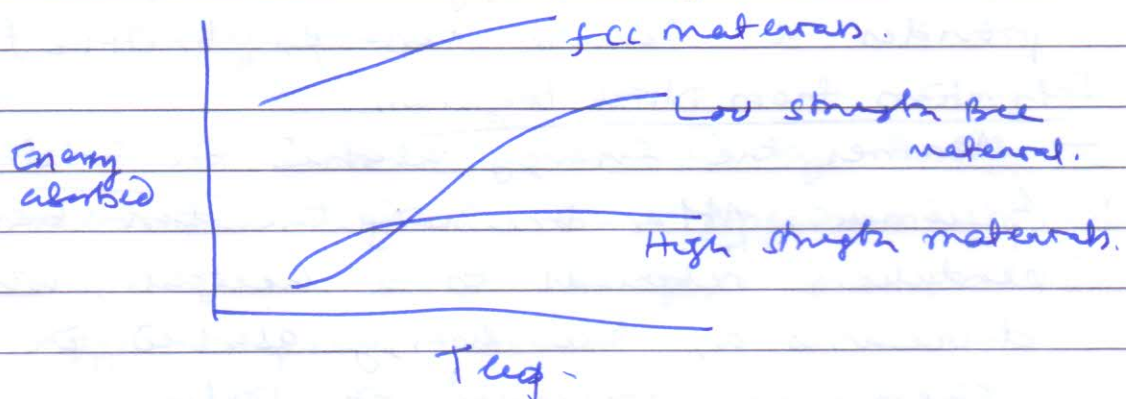
- Determination of DBTT.

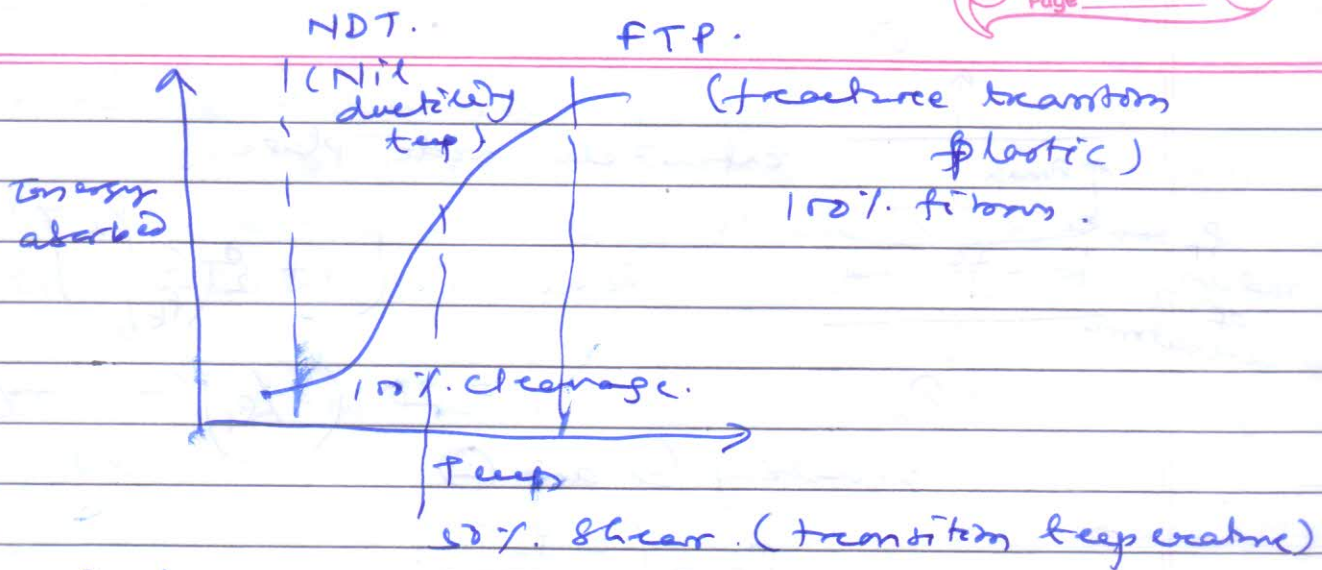


Steel 'A' shows higher notch toughness, yet its transition temperature is higher than that of Steel B. The material with the lowest transition temp is preferred.

Advantages of Charpy test

- Relatively simple, cheap and uses small test specimen.
- Comparison of alloys and their heat-treatment.
- Used for quality control.





factors affecting Notch toughness

- Chemical Composition.
 - physical properties (Hardness, microstructure, grain size, homogeneity, method of fabrication, Rolling direction, Hot or cold working temp).
- Chemical {
- Alloying elements
 - Gas content
 - Impurities.

fracture toughness

~~it was observed that theoretical cohesive strength is~~

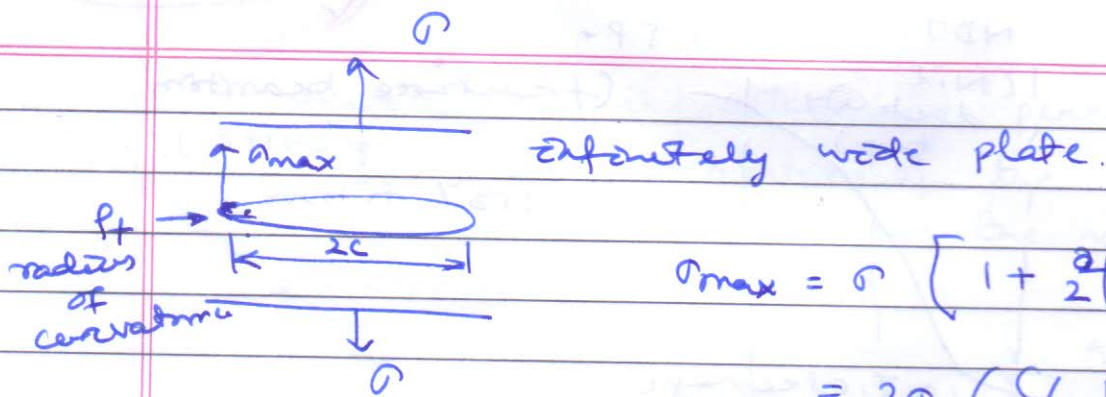
The investigations of brittle fractures by notch toughness tests led to development of new methods of notch toughness. But none of the methods gave rise to material constant. These parameters are affected by the size, shape, and notch. So a new parameter was introduced - called fracture toughness. This is based on the fracture mechanics approach.

Theoretical Cohesive strength of material

$$\sigma_{max} = \left(\frac{E \gamma_s}{a_0} \right)^{1/2} \quad \text{--- (1)}$$

fracture strength are 10 to 1000 times lower than the theoretical value.

Due to the presence of cracks are inevitable.



$$\sigma_{max} = \sigma \left[1 + 2 \left(\frac{c}{r_t} \right)^{1/2} \right]$$

$$= 2\sigma \left(\frac{c}{r_t} \right)^{1/2} \quad \text{--- (1)}$$

Equating (1) and (2)

$$\sigma_f = \left(\frac{E \gamma_s r_t}{4ac} \right)$$

Assuming the sharpest possible crack

$$r_t = a_0$$

$$\Rightarrow \sigma_f = \left(\frac{E \gamma_s}{4c} \right)^{1/2}$$

The first explanation of discrepancy between observed fracture strength of crystals and the theoretical cohesive strength was proposed by Griffith.

$$\sigma = \left(\frac{2E\gamma_c}{\pi c} \right)^{1/2}$$

For a brittle plate (brittle)

$$\sigma_f = \left[\frac{2E\gamma_c}{(1-\nu^2)\pi c} \right]^{1/2}$$

For brittle plate (Elastic + Plastic)

$$\sigma_f = \left[\frac{2E(\gamma_s + \gamma_p)}{\pi c} \right]^{1/2}$$

$$= \left[\frac{E\gamma_p}{c} \right]^{1/2}$$

fracture mechanics approach

It provides a quantitative framework for evaluating structural reliability in terms of applied stress, crack length, and stress intensity factor.

The LEFM approach is based on 3 major assumptions.

- Cracks and similar flaws are inherently present in parts

- A crack is a flat, internal free surface in a linear elastic stress field. (LEFM)

- The quantity of stored energy released from a crackling specimen or part during crack propagation is a basic material property, independent of specimen or part size

$$\sigma_y = \frac{k}{\sqrt{2\pi r}} \cos \theta/2 \left(1 + \sin \theta/2 \cdot \sin^3 \theta/2 \right)$$

$$\sigma_x = \frac{k}{\sqrt{2\pi r}} \cos \theta/2 \left(1 - \sin \theta/2 \cdot \sin^3 \theta/2 \right)$$

σ_y is the stress perpendicular to the crack plane.

σ_x is the stress perpendicular to the crack tip.

Stress intensity factor

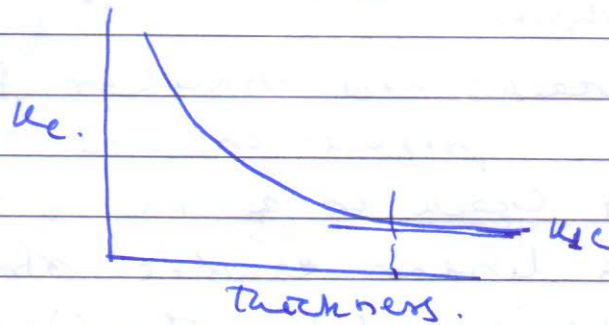
Rapid crack propagation is controlled by a material constant, called ^{critical} stress intensity factor (K_{IC}). It is that value of stress-intensity factor, k at which crack propagation becomes rapid. The greater the value of K_{IC} , greater the resistance to brittle fracture.

It is also called plane strain fracture toughness

$$K_{IC}^2 = E G_c \quad \text{Unit of } K_{IC} = \text{MPa}\sqrt{\text{m}}$$

G_c is the energy required for rapid crack propagation.

or Critical strain energy release rate for unstable crack propagation. (MJ/m^2)

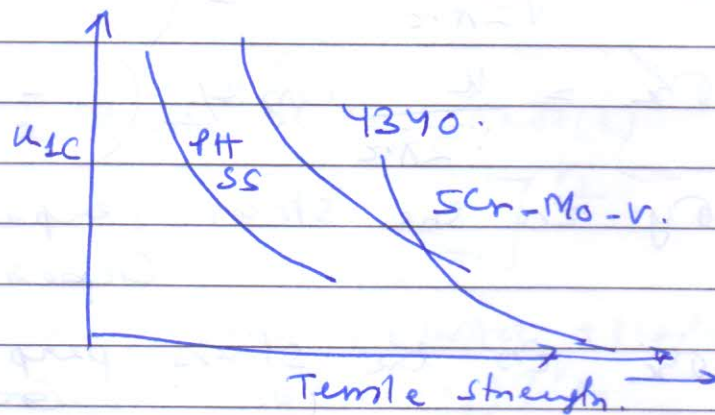


K_{IC} - Plane strain fracture toughness. is independent of thickness.

K_{IC} is determined by bend specimens.

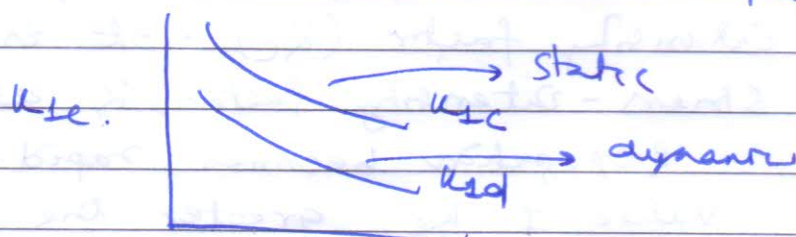
factor of safety (and) CT specimens. (Compact-tension)

K_{IC} vs Effect of Material Strength - If strength (TS) increases, the fracture toughness decreases.



2. Effect of loading rate.

$\frac{dK}{dt}$ should be in the range 0.55 to 2.75 $\text{MPa}\sqrt{\text{m}}/\text{s}^{-1/4}$



Q.1. A structural component of an engineering design loaded in plane strain condition must support 250 MPa load in tension. If the alloy used for this application is to be used without fail, calculate the length of the largest internal flaw that the alloy can withstand.

$$0.2\% \text{ Proof stress } (\sigma_{0.2}) = 503 \text{ MPa}$$

$$\text{Tensile strength (MPa)} = 572$$

$$\text{Shear strength (MPa)} = 331$$

$$\% \text{ elongation} = 11$$

$$\text{Vickers hardness (HV)} = 175$$

$$K_{Ic} = \text{MPa}\sqrt{\text{m}} = 25$$

$$\text{Fatigue strength } \sigma_f \text{ (MPa)} = 159.$$

Q.2. Discuss List the fundamental sources of failure. Discuss how the imperfections in materials, as a source of failure.

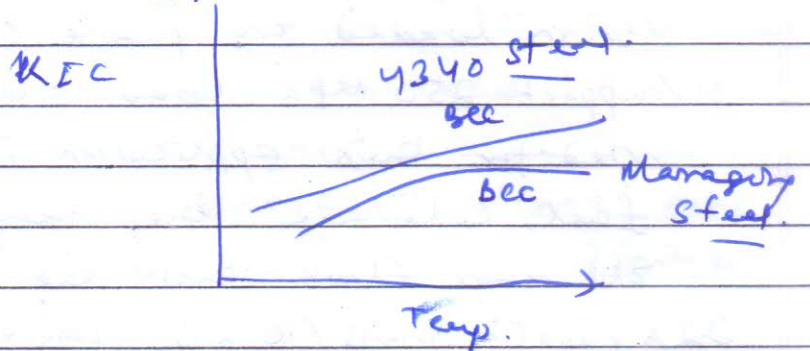
Q.3. Discuss the guidelines to reach the root cause of failure.

Q.4. What is failure analysis. What are the conditions to know whether a component has failed.

Q.5. Thin walled pressure vessels are to be made either from material A or Material B. If the applied stress is 1.2 times higher than when material 'A' is used and $(K_{Ic})_A = 0.6(K_{Ic})_B$ which material is to be used for pressure vessel. Explain with crack length calculation.

Q.6.

3. Effect of test temp.



Reverse trend is shown by Al alloys.

High temp strength and low

5. High and low temperature fractures.

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S.M.
Handbook

Mechanisms of and the influence of structural and environmental parameters on fracture.
Identification of types of fractures.

- Ductile & brittle.
- Fatigue fractures.
- Stress Corrosion Cracking.

Classification: (Macroscopic scale)

- Ductile & brittle
- Fatigue
- ~~Stress corrosion cracking~~ ^{combined effects of} stress and environment.

Assignment

- SCC
- Liquid metal embrittlement
- Interstitial embrittlement
- Corrosion fatigue.

Same as fatigue.

Ductile

Gross plastic deformation.
gray, fibrous.
in appearance.

↓

Microscopic:

flat faced (square) Shear-face (slant).

→ produced under plane strain condition of ductile material.

— perpendicular to the direction of loading.

— Equiaxed dimples.

Brittle

- 1 Less plastic deformation
 - 2 Bright, granular appearance.
 - 3 - Produced under plane-strain
- plane stress. (thin section)
- 45° to the tensile axis.
- Elongated dimples.

4. perpendicular to the direction of loading.

5. Chevron pattern may be present at the origin.

6. Ductile granular or transgranular.

Trans granular fractures are by microvoid coalescence cleavage, quasi cleavage, Ductile granular fracture are by GB separation, with or without microvoid coalescence.

Examples

Leaf Springs
A/Cr wheels

→ Fatigue fractures

Occurs due to the combined action of cyclic stress, tensile stress and plastic strain.

↳ Result from cyclic loading.

↳ Appear brittle in microscopic scale.

↳ Beach marks.

One microscopic scale it consists of 3 parts.

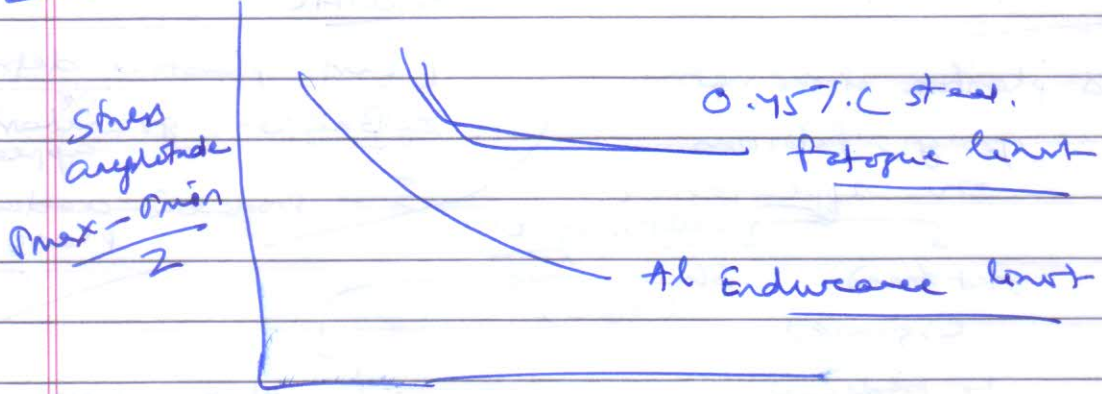
3 stages.

- Initial fatigue damage leading to crack initiation

- Crack propagation until the remaining uncracked cross section of a part

- Final sudden fracture.

Types

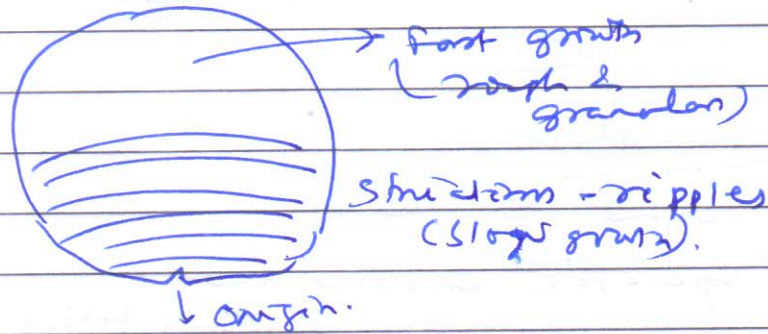
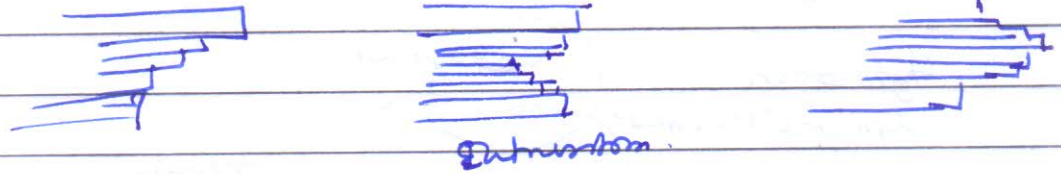


Initiation

$\sigma_{fatigue} < \sigma_{UTS} \sigma_{ys}$
(Fatigue strength)

↳ Occurs due to microscopic plastic deformation in localized scale during stress reversals.

~~Area~~ Slip starts at the SFC

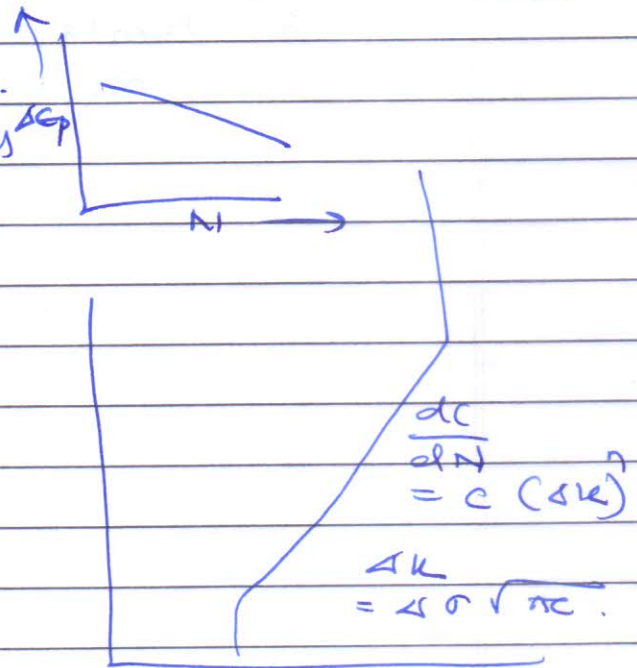


Low cycle fatigue

High cycle fatigue

- $> 10^5$
- thermal origin.
- pressure vessels

Crack growth rate.



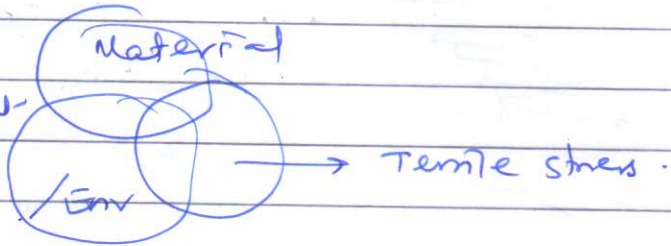
Factors

- frequency - has small effect
- form of stress - No effect
- Environment - needed effect.
- thickness - thinner materials show low crack growth rate.
- Temperature $\rightarrow \uparrow \bullet$

- Prevention - shot peening - compressive stress.
- Carburizing & Nitriding.
 - fine grain size.
 - Avoidance of stress concentration (highly polished SFC)

Stress Corrosion Cracking

Hydrogen embrittlement



Characteristics

1. Found in alloys
2. Specific environment
mild steel - conc HNO_3 , OH^-
stainless steel - Cl^-
3. Tensile component of stress
& Residual stresses.

→ Season Cracking of Brass
Cold drawn cartridge
Cases of α -brass in Ammonia.
→ Caustic embrittlement
of boilers.

Boiler water treated
with NaOH to
control pH.

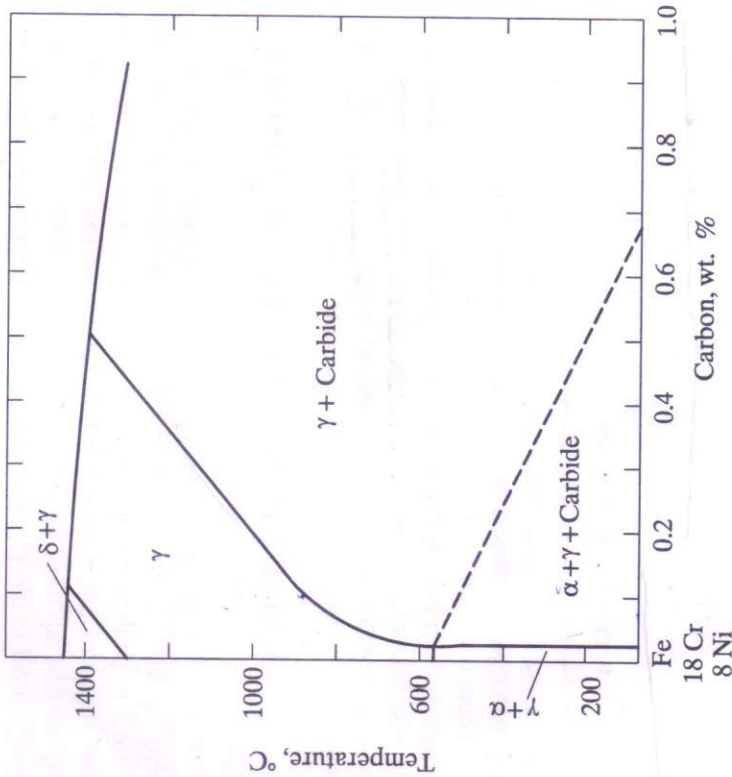


Figure 6.11 The pseudo-binary phase diagram for an 18/8 stainless steel as a function of carbon content. The precipitation of the α phase below the dotted line rarely occurs.

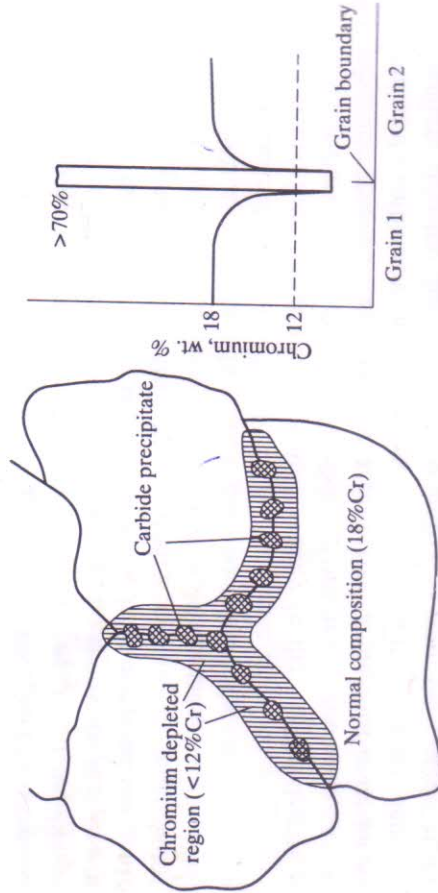


Figure 6.12 Chromium carbide precipitation at a grain boundary and the composition profile near the boundary.

1. The steel is reheated into the single austenitic region, Fig. 6.11, and then quenched. The carbides go into solution during austenitizing and have no time to reprecipitate during quenching. This procedure cannot be adopted, if the steel is to be used in service at temperatures between 500 and 800°C, as grain boundary precipitation will again occur during service.

2. A low carbon grade of stainless steel such as 304L or 316L can be used. In these grades, the carbon content is limited to a maximum of 0.03%. So, there is no precipitation of carbides, note the solubility of carbon in austenite in the phase diagram in Fig. 6.11.

3. The stabilized grades 321 and 347 can be used. In type 321, titanium greater than 5 times the wt% C is added. In type 347, niobium greater than 10 times the wt% C is present. Both titanium and niobium form more stable carbides than chromium, recall the carbide stability scale given on p. 107. So, carbon combines preferentially with titanium or niobium. Chromium remains in solution and sensitization does not occur.

The above description is valid for austenitic stainless steels. Because of the low solubility of carbon in ferrite, ferritic stainless steels sensitize much more rapidly and at lower temperatures. Here, the intergranular corrosion cannot be prevented, by a solutionizing treatment and water quenching. They must be annealed at about 800°C to replenish the Cr-depleted regions near the grain boundaries by diffusion of Cr from the interior of the grains. Reducing C to 0.03% is also not applicable here. Duplex stainless steels, which contain meta-stable ferrite within the austenite grains, are known to be remarkably resistant to intergranular corrosion.

304 - 18-8 - 0.08
 304L - - - 0.03
 316 - - - 2 Nb
 321 - - - 5 Ti
 347 - - - 10 Nb

6.9 STRESS CORROSION CRACKING

Stress corrosion cracking (SCC) refers to the cracking of a material under static load by the combined action of a stress and a corrosive environment. The related phenomenon of hydrogen cracking or hydrogen embrittlement is due to the absorption of atomic hydrogen into the metal.

General characteristics The experimentally known characteristics of stress corrosion cracking are summarized below.

1. Stress corrosion cracking is found in alloys and not in pure metals. However, there is some uncertainty as to what level of purity can be classified as "pure"!
2. SCC occurs only in a specific environment for a given alloy. Some examples of environments and the corresponding alloys are given below.

A sensitized steel can be restored to its original state of excellent corrosion resistance in the following ways.

Alloy	Environment
Mild steel	Conc. NO_3^- , OH^-
High strength steel (T.S. > 1200 MPa)	H_2O
Stainless steel	Cl^-
α -brass	Traces of NH_3 or NH_4^+
Aluminium alloys	H_2O , NaCl solutions

- The presence of a *tensile component* of stress is necessary. The stress may be externally applied. Or it can be residual stresses in the alloy. Early examples of SCC were due to residual stresses: the *season cracking* of brass and the *caustic embrittlement* of boilers. When cold drawn cartridge cases of α -brass are exposed to traces of ammonia in the atmosphere, they crack spontaneously. When the cold worked brass is given a stress-relieving anneal at 250–300°C, the susceptibility to SCC disappears. Boiler water is treated with NaOH to control *pH*. In the crevices between bolts and the boiler plates, evaporation results in a concentration of OH^- ions, leading to SCC of the bolts having residual stresses induced by plastic deformation during tightening. Welded boilers are also prone to SCC, in the presence of residual welding stresses.
- The cracks in SCC move on a very narrow front. In hydrogen cracking, the cracks are known as *hairline cracks* often difficult to identify. If cracking is due to anodic dissolution at the crack tip, extremely localized corrosion must be taking place.
- The crack path is often *intergranular* in SCC, Fig. 6.13. Transgranular crack propagation, i.e. propagation across the grains, has also been found in some cases.

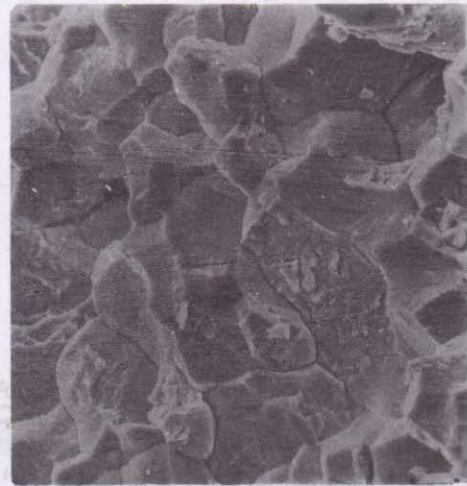


Figure 6.13 Intergranular fracture in stress corrosion cracking.

- Many of the alloys which are susceptible to SCC form passive films. Examples are aluminium alloys and stainless steels. However, this is not always the case.

- Cracking consists of two stages: (i) *initiation* and (ii) *propagation*. Titanium alloys are immune to crack initiation in chloride solutions. But when the alloys are precracked, crack propagation is found to occur.

Mechanisms of crack growth In the electrochemical mechanism, corrosion by anodic dissolution is believed to occur at the crack tip. In alloys that form a passive film, the crack tip is not covered by the film, because the plastic deformation at the tip exposes fresh metal or the stress concentration at the tip is sufficient to break the brittle oxide film. An active-passive cell is set up between the tip and the crack faces that are covered with the film. Only a very small area of the tip is exposed, resulting in a high current density at the tip. Intergranular corrosion is due to the galvanic cells set up at the grain boundaries by segregation or precipitation. In this mechanism, cathodic protection should be effective in preventing SCC, as is found in some cases.

In hydrogen cracking, the adsorption of atomic hydrogen at the crack tip faces can lower the surface energy γ in the Griffith equation (5.6). This explanation is probably valid for the time-dependent cracking observed in non-metallic materials, where an electrochemical mechanism is not possible. In ductile alloys, γ_p is much larger than γ and an adsorption mechanism is unlikely. Hydrogen atoms presumably diffuse into the metal near the crack tip and *lock up the mobile dislocations* thus preventing plastic flow. The yield strength increases locally and the lack of plastic flow at the tip causes embrittlement.

Alternatively, supersaturated hydrogen atoms can precipitate as H_2 molecules at a crack nucleus or interface. The stresses from the build-up of hydrogen pressure can induce the nucleation and/or growth of cracks. At room temperature, hydrogen being a small interstitial atom can diffuse over a number of interatomic distances in a matter of seconds. A high hydrogen overvoltage slows down the combination of hydrogen atoms into molecular hydrogen, thus facilitating their inward diffusion. Sulphur containing oil increases the hydrogen overvoltage and thereby causes embrittlement of oil pipelines.

The stress intensity factor K_{Isc} The time dependent fracture under a corrosive environment has been studied experimentally by measuring the stress intensity factor K_I . After precracking, a cantilever specimen that is surrounded by the corrosive environment is subjected to a constant load. The measured time for fracture is plotted in Fig. 6.14 as a function of K_I . If the applied stress corresponds to K_{Ic} , fast fracture occurs instantly. When $K_I < K_{Ic}$, fracture occurs after a delay period, during which the crack grows slowly. When the stress intensity factor is below a critical value known as K_{Isc} , no fracture occurs even after prolonged holding, see Fig. 6.14.

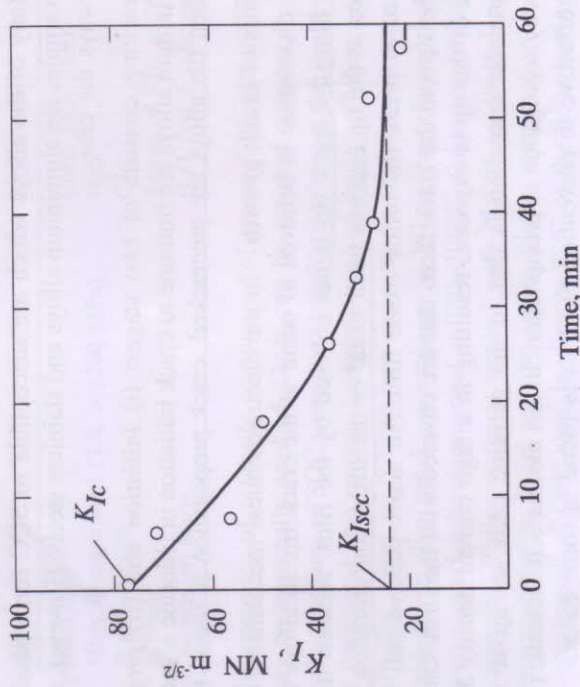


Figure 6.14 The critical stress intensity factor decreases with increasing time in a corrosive environment, reaching a constant value of K_{Isc} below which no cracking occurs.

Figure 6.15 shows the measured rates of crack growth as a function of K_I . Below K_{Isc} the crack growth rate is zero. Above K_{Isc} in region II, the crack grows at a constant rate independent of K_I . The growth rate is probably controlled by an electrochemical process in this region. Region III corresponds to fast growth that culminates in fracture.

The energy balance for crack growth in the Griffith equation (5.6) needs to be modified to take into account the electrochemical energy released at the crack tip due to anodic dissolution. For unit extension of the crack, we can write the condition for spontaneous growth as:

$$\underbrace{\text{surface energy} + \text{plastic work}}_{\text{of the crack at the tip}} = \underbrace{\text{elastic energy} + \text{electrochemical energy released}}_{2\gamma_p} + (nF\rho\delta M)\epsilon$$

where σ is the tensile stress,
 c is half-length of the crack,
 Y is Young's modulus of the alloy,
 n is the number of electrons taking part in the anodic reaction,
 F is Faraday's constant,
 ρ is the density of the alloy,
 δ is the opening at the crack tip,

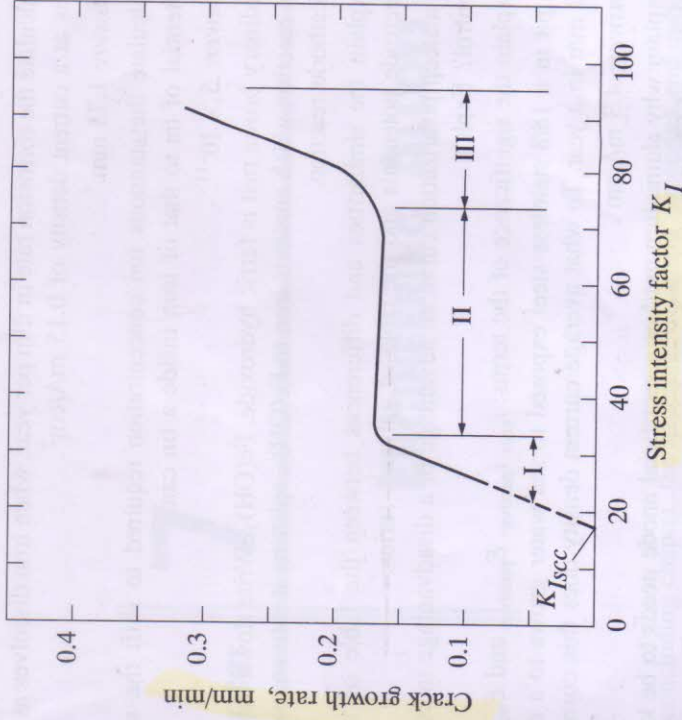


Figure 6.15 The crack growth rate in a Ni-Cr-Mo steel in a NaCl environment as a function of the stress intensity factor, depicting the three stages.

Rewriting the critical condition in terms of K_{Isc} we have

$$K_{Isc} = [2\gamma_p Y - (nF\rho\delta M)\epsilon]^{1/2} \quad (6.6)$$

Equation (6.6) shows that K_{Isc} decreases with an increase in yield strength (which decreases γ_p) and with an increase in the corrosion tendency as given by ϵ .

SUGGESTED READINGS

Jones, D.A., *Principles and Prevention of Corrosion*, Prentice Hall, New Jersey (1996).

Uhlig, H.H., *Corrosion and Corrosion Control*, John Wiley, New York (1971).

QUESTIONS

1. Calculate the amount of zinc (anode) that corrodes when a current of